# **Realizing** *p***-type NbCoSn** half-Heusler compounds with enhanced thermoelectric performance via Sc substitution

Ruijuan Yan<sup>1</sup>, Wenjie Xie<sup>1</sup>, Benjamin Balke<sup>2</sup>, Guoxing Chen<sup>1</sup>, Anke Weidenkaff<sup>1,2</sup>

<sup>1</sup> Department of Materials Science, Technische Universität Darmstadt, Darmstadt, Germany

<sup>2</sup> Fraunhofer Research Institution for Materials Recycling and Resource Strategies IWKS, Alzenau, Germany

## Abstract

N-type half-Heusler NbCoSn is a promising thermoelectric material due to favourable electronic properties. It has attracted much attention for thermoelectric applications while the desired NbCoSn counterpart shows poor thermoelectric *p*-type performance. In this work, *p*-type NbCoSn has been obtained using Sc substitution at the Nb site, and their thermoelectric properties were investigated. Of all samples, Nb<sub>0.95</sub>Sc<sub>0.05</sub>CoSn compound shows a maximum power factor of 0.54 mW/mK<sup>2</sup> which is the highest among the previously reported values of *p*-

# **Electrical properties**





type NbCoSn. With the suppression of thermal conductivity, ptype  $Nb_{0.95}Sc_{0.05}CoSn$  compound shows the highest measured figure of merit ZT = 0.13 at 879 K.



Figure 1 Crystal structure of NbCoSn and schematic illustration of Sc substitution

#### **Phase characterization**

> All samples show the cubic MgAgAs crystal structure with some minor Nb<sub>3</sub>Sn impurities.

> The cell parameter increases with Sc concentration because of the ionic radius difference between  $Sc^{3+}$  (0.87 Å) and  $Nb^{5+}$  (0.74 Å).



Figure 4 Temperature dependence of the electrical transport properties of Nb<sub>1-z</sub>Sc<sub>z</sub>CoSn (a) electrical conductivity (b)  $\ln \sigma vs$ . 1000/T plot.



(b) of Nb<sub>1-z</sub>Sc<sub>z</sub>CoSn samples and the power factor of Nb<sub>0.8</sub>Zr<sub>0.2</sub>CoSn as comparison [2].



 $Nb^{5+}=Sc^{3+}+2h^{+}$ 

≻The carrier concentration trend matches with the  $\sigma$  variation.  $\sim$  The electrons are dominant (Sc < 0.05) and with increasing Sc content, n decreases.  $\blacktriangleright$  When Sc  $\geq$  0.05, the dominant carriers are holes and the cont-

Figure 2 (a) PXRD patterns and (b) cell parameters of Nb<sub>1-7</sub>Sc<sub>7</sub>CoSn samples.

There is no obvious phase segregation and the elements almost distribute uniformly in  $Nb_{0.95}Sc_{0.05}CoSn$  sample.



Figure 6 The carrier concentration and carrier mobility of Nb<sub>1-7</sub>Sc<sub>7</sub>CoSn samples at room temperature (a) *n*-type, (b) *p*-type.

### **Thermal properties**



 $\triangleright$  After substituting Sc, the  $\kappa$  decreases dramatically from 11.14 W/ mK to 4.25 W/mK at room temperature due to the reduction of  $\kappa_{\rm L}$ , which originates from the intensive point defect scattering ( mass fluctuation and strain field fluctuation).

Figure 7 The temperature dependence of total thermal conductivity  $\kappa$  (a) and lattice thermal conductivity  $\kappa_{\rm L}$  (b).

ZT





Figure 3 The secondary electron image and the elemental distribution mappings of Nb<sub>0.95</sub>Sc<sub>0.05</sub>CoSn.

The value of ZT improved to 0.13 at 879K in *p*-type  $Nb_{0.95}Sc_{0.05}CoSn$  sample, which is much higher than that of  $Nb_{0.8}Zr_{0.2}CoSn$ .

Figure 8 The figure of merit ZT for p-type NbCoSn and Nb<sub>0.8</sub>Zr<sub>0.2</sub>CoSn and Nb<sub>0.8</sub>Zr<sub>0.2</sub>CoSn as comparison [2].

### Conclusions

- $\succ$  Sc is substituted at Nb site successfully and changes the original *n*-type NbCoSn to *p*-type semiconductor, indicating Sc is an efficient *p*-type dopant.
- > Substituting Sc generates point defects and reduces lattice thermal conductivity.
- $\succ$  The highest ZT value of 0.13 is achieved in p-type Nb<sub>0.95</sub>Sc<sub>0.05</sub>CoSn.

# References

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# Acknowledgement

This work is supported by DFG project BA 4171/4-1. Besides, thanks for help from Dr. Wenjie Xie, Dr. Benjamin Balke and Dr. Guoxing Chen.

